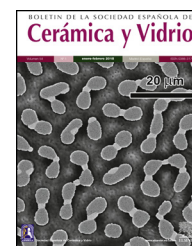




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## Synthesis of Fe, Mn and Cu modified TiO<sub>2</sub> photocatalysts for photodegradation of Orange II

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### ABSTRACT

The residual dyes from different sources are considered a wide variety of organic pollutants introduced into the natural water resources. One of the main sources with severe pollution problems worldwide is the textile industry and its dye-containing wastewaters. The aim of this study was the photocatalytic degradation of the azoic dye “Orange II” using Mn, Cu and Fe modified titania. The catalysts were prepared by two different methods, sol-gel and an impregnation method in a commercial titania (Degussa P25), after drying the samples were calcined at 400 °C. The samples were characterized by BET method for surface area measurement ( $S_{\text{BET}}$ ), XRD, Raman spectroscopy, XPS, UV-vis spectroscopy by diffuse reflectance technique and its photocatalytic activity was evaluated in the degradation of the azo dye Orange II in aqueous solutions using the Langmuir-Hinshelwood model. The catalysts prepared by sol-gel method exhibit a larger surface area than the prepared by impregnation and the introduction of the oxides leads to a slight reduction of  $S_{\text{BET}}$ . The sol-gel samples show lower photoactivity than the impregnated samples due to the presence of both anatase and rutile phases in these powders, the phase junction anatase-rutile facilitate transfer of the photogenerated electron from the conduction band of the rutile phase to the trapping sites on the anatase surface and prevents the electron-hole recombination and allows the holes generated to move to the surface of the catalyst. XRD, Raman, XPS and UV-Vis data show that the metallic cations are incorporated in the titania lattice in greater extend in SG samples while remain dispersed on the surface in impregnated samples. Although the entrance of metallic cations in solid solution enhance the photocatalytic activity due to the enhanced formation of electrons-hole pairs, at a high doping content, a large number of

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structural defects are induced serving as a recombination center of electron–hole pairs and sol–gel samples show higher photoactivity than impregnated samples, except in the case of Fe modified samples that shows the higher red shift with  $E_g = 1.8$  eV and compensate the induced recombination center of electron–hole pairs.

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## Síntesis de fotocatalizadores de TiO<sub>2</sub> modificado con cationes de Fe, Mn y Cu para la degradación de Naranja II

### R E S U M E N

*Palabras clave:*  
Fotocatálisis  
Óxido de titanio  
Sol-gel  
Impregnación  
Naranja II

Los tintes orgánicos residuales de la industria textil suponen una amplia variedad de contaminantes orgánicos introducidos en los recursos hídricos naturales. Una de las fuentes principales con graves problemas de contaminación en todo el mundo es la industria textil y las aguas residuales que contienen estos tintes. El objetivo de este estudio fue la degradación fotocatalítica del tinte azoico «Naranja II» utilizando óxido de titanio modificado con Mn, Cu y Fe. Los catalizadores se prepararon mediante dos métodos diferentes, sol-gel y un método de impregnación en un óxido de titanio comercial (Degussa P25), después de secar las muestras se calcinaron a 400 °C. Las muestras se caracterizaron por el método BET para la medida de la superficie específica ( $S_{BET}$ ), XRD, espectroscopia Raman, XPS, espectroscopia UV-vis mediante la técnica de reflectancia difusa y se evaluó su actividad fotocatalítica en la degradación del colorante azoico Orange II en soluciones acuosas utilizando el modelo de Langmuir-Hinshelwood. Los catalizadores preparados por el método sol-gel presentan una superficie específica mayor que la preparada por impregnación y la introducción de los óxidos conduce a una ligera reducción de la  $S_{BET}$ . Las muestras de sol-gel muestran una fotoactividad más baja que las muestras impregnadas debido a la presencia de anatasa y rutilo en estos polvos, la unión de fase anatasa-rutilo facilita la transferencia del electrón fotogenerado de la banda de conducción de la fase rutilo a los sitios de captura en la superficie de anatasa y evita la recombinación de los pares electrón-hueco permitiendo que los huecos generados se muevan hacia la superficie del catalizador. Los datos de XRD, Raman, XPS y UV-Vis muestran que los cationes metálicos se incorporan en la red de óxido de titanio con mayor amplitud en las muestras SG, mientras que permanecen dispersos en la superficie en muestras impregnadas. Aunque la entrada de cationes metálicos en solución sólida mejora la actividad fotocatalítica debido a la más fácil producción de pares electrón-hueco, con un alto contenido de dopante, se inducen una gran cantidad de defectos estructurales que sirven como centro de recombinación de pares electrón-hueco. Las muestras sol-gel muestran una fotoactividad más alta que las muestras impregnadas, excepto en el caso de muestras modificadas con Fe que muestran un mayor desplazamiento al rojo con  $E_g = 1.8$  eV y compensan el centro de recombinación inducida de pares electrón-hueco.

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## Introduction

Since the industrial revolution, the manufacturing processes have become more efficient and productive and science has become much more advanced, thus leading to an improvement in the quality of human life. On the other hand, all the advancement and development witnessed have also brought with them a wide spectrum of problems, water pollution being one of them. The wastewater poses serious threats to human health and the environment. Concretely, the textile industry is one of the huge sources of water pollution. This industry generates highly toxic chemical concentration at every step in the production process, as the dyes are occupying an important class of these pollutants. The synthetic dyes are toxic to

aquatic life, reduce photosynthesis, carcinogenic, mutagenic and some of them make allergy, dermatitis and skin irritation to human [1,2]. Also, the chemical structures of dyes vary enormously, and some have complicated aromatic structures that resist degradation in conventional wastewater treatment processes [3–5]. In this regard, various treatment techniques have been developed over the last decade, among of them, heterogeneous photocatalytic.

The photocatalytic oxidation based on semiconductor material is greatly efficient and environmentally friendly processes for various wastewater treatments. Accordingly, in the present study TiO<sub>2</sub>-based catalysts were synthesized for the degradation of the azoic dye “Orange II”. Titania (TiO<sub>2</sub>) is widely used for the removal of highly toxic and non-biodegradable pollutants frequently presents in wastewater

by means of photocatalysis. Sol-gel method has been widely used to synthesize versatile titania in powder or thin films coatings photocatalyst [6–10]. However, the photocatalytic activity of TiO<sub>2</sub> is limited, due to its wide band gap property of absorbance only in short wavelength. Recently, several studies have shown that the presence of some transition metals enhances the photoactivity of titania in the degradation of organic pollutants and that the photoactivity of these materials depends strongly on the character and the concentration of the employed dopant [6–11].

The textile industry uses more than 30,000 different compounds, of which more than 10,000 are dyes. Among the dyes, the azoics are the most common since they account for 60–70% of the total dyes used. Its degradation by conventional methods is complicated among other reasons because of its low water solubility. Although conventional treatments with oxidants often allow the complete discoloration of textile effluents, however, they do not achieve complete mineralization. Photocatalysis allows completely discoloring and mineralizing this type of dyes [12]. The mechanism of degradation of Orange II is well described in the literature. The oxidative attack on the azo group produces benzene sulfonate and naphthoquinone in the first phase. Likewise, more complete studies identify up to 22 transformation intermediates of Orange II that include 2-naphthol, 2-hydroxy-1,4-naphthoquinone, small aromatics such as phthalic acid and phthalimide and aliphatic acids such as fumaric, succinic, maleic and Malonic [13,14].

Although titania metal doping and the sol-gel route of synthesis are well known on Orange II photocatalytic degradation, usually the method is not compared with the introduction of metal doping to a recognized commercial photocatalyst using similar characterization techniques. In this study the sol-gel method is compared with the introduction of metal doping to Degussa P25 titania by an impregnation method in order to evaluate the efficiency of the method. In this sense, this paper focuses on the photocatalytic activity of pure and modified titania in the removal of Orange II (OII) in aqueous solution. Concretely, the aim of the present study is to prepare Mn, Cu and Fe modified titania, following two different methods: sol-gel and impregnation, in order to investigate the influence of the preparation method on their physico-chemical properties and their activity in photocatalytic oxidation of OII.

## Experimental

### Preparation of catalysts

Two types of TiO<sub>2</sub> were used in this study: the commercial Degussa P25 titania (TiO<sub>2</sub>-P25) and another one prepared by a sol-gel procedure (TiO<sub>2</sub>-SG).

TiO<sub>2</sub>-SG was prepared from titanium (IV) isopropoxyde according to the following molar ratios: Ti(C<sub>12</sub>H<sub>28</sub>O<sub>4</sub>):6 C<sub>2</sub>H<sub>5</sub>OH:16 H<sub>2</sub>O. Thus, the solution of water and ethanol was prepared. Then, an amount of Ti(C<sub>12</sub>H<sub>28</sub>O<sub>4</sub>) was added dropwise using a syringe to avoid absorption of moisture from the atmosphere, while stirring continuously until gel formation. This gel was aged at constant temperature (70 °C), during 4 h. The resulting gels were dried at 110 °C for 24 h and calcined

at 400 °C for 12 h, using a heating rate of 10 °C min<sup>-1</sup>. It is worth noting that the pH was adjusted to 3 by adding nitric acid (HNO<sub>3</sub>).

Mn, Cu and Fe modified TiO<sub>2</sub> were prepared by two different methods: sol-gel and impregnation, in order to evaluate the effect of the preparation method on the chemical and textural properties of the elaborated catalysts.

Mn/TiO<sub>2</sub>-SG, Cu/TiO<sub>2</sub>-SG, Fe/TiO<sub>2</sub>-SG samples prepared by a sol-gel process were obtained adding, respectively, aqueous solutions of Mn(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O, Cu(NO<sub>3</sub>)<sub>2</sub>·2.5H<sub>2</sub>O and Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O to the Ti(C<sub>12</sub>H<sub>28</sub>O<sub>4</sub>) in H<sub>2</sub>O/HNO<sub>3</sub> solution. The metal content in the samples was of 5 wt.%. The rest of the process is the same used for the preparation of the pure TiO<sub>2</sub>-SG. The obtained catalysts were calcined at 400 °C for 12 h (10 °C min<sup>-1</sup>).

Mn/TiO<sub>2</sub>-I, Cu/TiO<sub>2</sub>-I and Fe/TiO<sub>2</sub>-I catalysts were synthesized by the impregnation method. 5 g of a commercial titania (Degussa P25, 83% anatase and 17% rutile) was dispersed in 100 ml of 0.04 mol/L solutions of Mn(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O, Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O and Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O respectively. The metal loading was of 5 wt.%. The suspension was kept under stirring during 24 h at room temperature. Then, the samples were dried overnight at 110 °C and calcined at 400 °C for 12 h, with a heating rate of 10 °C min<sup>-1</sup>.

### Catalysts characterization

The textural properties of the catalysts have been determined by N<sub>2</sub> adsorption at -196 °C, on a fully automated Micromeritics 3-Flex Surface Characterization Analyzer. The specific surface area (S<sub>BET</sub>) was obtained by applying the BET method to the experimental N<sub>2</sub> adsorption data.

The X-ray diffraction patterns of the catalysts were recorded on an X'Pert Pro MPD diffractometer from Panalytical, powered by a PW3040/60 X-ray generator and fitted with a Cu K<sub>α</sub> radiation source (wavelength K<sub>α1</sub> = 1.54060 Å, K<sub>α2</sub> = 1.54443 Å), a step size of 0.06° 2θ/s and time step of 120 s and a 2θ range of 5–90° which operated at 45 kV and 40 mA. The lattice parameters (*a* and *c*) can be equated for the tetragonal system (*a* = *b* ≠ *c*) as follows:

$$\frac{1}{d_{hkl}^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$

where *d* is the interplanar distance and *h*, *k* and *l* are the Miller indices. The average crystal size for TiO<sub>2</sub> was estimated by application of the Scherrer equation [15] to the (1 0 1) anatase diffraction peak and (1 1 0) of rutile. Furthermore, the content of anatase and rutile was calculated according the Spurr-Myers equation [16].

Raman spectra were recorded on a FRA 106/S FT-Raman spectrometer, covering a spectral range of 36–3600 cm<sup>-1</sup> for stokes and 100–2000 cm<sup>-1</sup> for anti-stokes. This equipment has a near infrared excitation laser source (Nd-YAG) and a high sensitivity liquid N<sub>2</sub>-cooled Ge.

X-ray photoelectron spectroscopy was performed with a K-ALPHA spectrometer (Thermo Scientific). All spectra were collected using Al-K<sub>α</sub> radiation (1486.6 eV), monochromatized by a twin crystal monochromator, yielding a focused X-ray spot with a diameter of 400 nm, at 3 mA × 12 kV. The alpha hemispherical analyser was operated at the constant energy

mode with survey scan pass energies of 200 eV to measure the whole energy band and 50 eV in a narrow scan to selectively measure the particular elements. Charge compensation was achieved with the system flood gun that provides low energy electrons and low energy argon ions from a single source. The powder samples were pressed and mounted on the sample holder and placed in the vacuum chamber. Before recording the spectrum, the samples were maintained in the analysis chamber until a residual pressure of ca.  $5 \times 10^{-7}$  Nm<sup>-2</sup> was reached. The quantitative analysis was estimated by calculating the integral of each peak, after subtracting the S-shaped background, and by fitting the experimental curve to a combination of Lorentzian (30%) and Gaussian (70%) lines.

The UV–vis spectroscopy analysis was carried out in a JASCO V-670, UV–vis–NIR spectrophotometer equipped with an integrating sphere diffuse reflectance accessory, using BaSO<sub>4</sub> as reference scatter. The spectra were taken in the wavelength range of 200–800 nm. The band gap energy of catalysts was calculated by Tauc plot [17].

### Photocatalytic activity tests

The photoactivity of the samples has been evaluated in the degradation of the azo dye Orange II in aqueous solutions. The photocatalytic tests were carried out adding a catalyst loading of 0.5 g/L to an OII solution of  $0.6 \times 10^{-4}$  mol/L, which has been prepared dissolving an amount of C<sub>16</sub>H<sub>11</sub>N<sub>2</sub>NaO<sub>4</sub>S in a pH 7.42 phosphate buffer media (NaH<sub>2</sub>PO<sub>4</sub>, H<sub>2</sub>O 3.31 g and Na<sub>2</sub>HPO<sub>4</sub>, 7H<sub>2</sub>O 33.77 g solved in 1L of water). The UV irradiation source was a mercury lamp of 125 W emitting in the range 254–365 nm. Before measuring, the suspensions were first stirred in the dark during 15 min to reach sorption/desorption equilibrium. The evolution of reaction was followed taking samples every 15 min. Orange II concentrations in the different sample were determined by means of UV–vis spectroscopy at 485 nm. The commercial TiO<sub>2</sub> Degussa P25 was used as reference of comparison with the home prepared samples.

## Results and discussions

### Characterization

#### Textural properties

Table 1 reports the textural properties of all the samples, obtained from the N<sub>2</sub> isotherms at –196 °C (not shown). The

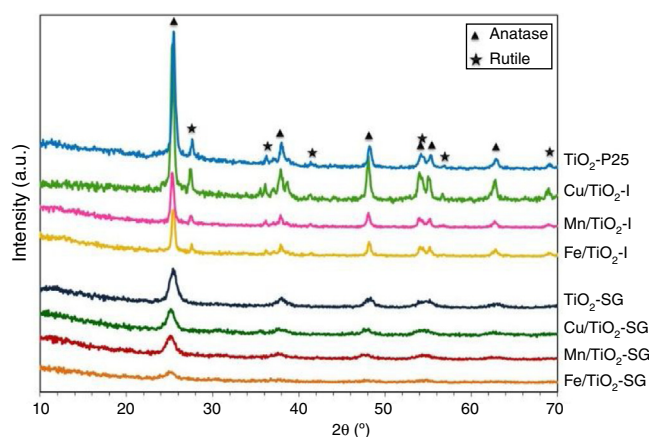


Fig. 1 – X-ray diffraction patterns of all the catalysts.

catalysts prepared by sol–gel method exhibit a larger surface area than their counterparts prepared by impregnation. Indeed, the surface area of all the sol–gel catalysts is almost three times higher than that found for the same samples prepared by impregnation. Moreover, it can be seen that the introduction of the oxides leads to a slight reduction of the surface area accompanied, as expected, with a decrease in the micro and mesopores volumes, can be due to a partial obstruction of TiO<sub>2</sub> pores by the metal particles [18,19].

#### X-ray diffraction

Fig. 1 shows the XRD patterns for the different catalysts. Commercial titania (TiO<sub>2</sub>-P25) pure and modified with Cu, Mn and Fe show the characteristic peaks of the anatase and the rutile phases, with a larger proportion of anatase [20–22]. However, the sol–gel catalysts show only broad and little intense bands attributed to the anatase phase [16]. For the modified catalysts, no metallic oxides diffraction peaks were observed, suggesting a good dispersion of the metal species on the catalyst surface.

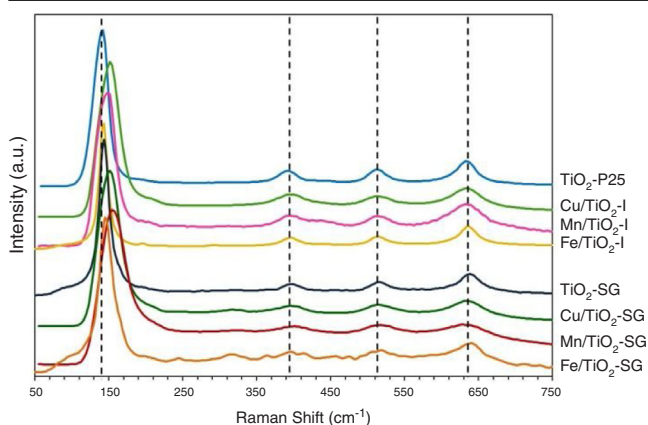
The TiO<sub>2</sub> crystallite sizes estimated from Scherrer's equation from the (1 0 1) and (1 1 0) reflections of anatase and rutile, respectively, are summarized in Table 2. It can be seen that the sol–gel method leads to catalysts with a much lower particle size of titania compared with that obtained for their counterparts prepared by impregnation. These catalysts show only the anatase phase with a small particle size which explains the high superficial area. On the other hand, the XRD diffractograms of all the modified samples, both those prepared by impregnation and sol–gel method, show a slight shift of the peaks toward lower 2θ values as compared with pure titania in the case of SG samples which can be associated with the incorporation of the large transition metal ions in the TiO<sub>2</sub> lattice generating some perturbation in the crystalline structure of the latter [23]. Furthermore, there is a clear variation of the parameters *a* and *c* of TiO<sub>2</sub> unit cell after the modification that are presented in Table 2. For SG samples *a* and *c* lattice parameters increase with the incorporation of the metallic cations into the titania network, producing changes in unit-cell parameters as a consequence of the different sizes of these cations and Ti<sup>4+</sup>. For impregnated samples I the change in lattice parameter is small and tending to lower values [23,24].

Table 1 – Textural properties of the catalysts.

Sample	S <sub>BET</sub> (m <sup>2</sup> g <sup>-1</sup> )	V <sub>T</sub> (cm <sup>3</sup> g <sup>-1</sup> )	V <sub>0</sub> (cm <sup>3</sup> g <sup>-1</sup> )	V <sub>meso</sub> (cm <sup>3</sup> g <sup>-1</sup> )
TiO <sub>2</sub> -P25	50	0.60	0.03	0.57
Cu/TiO <sub>2</sub> -I	38	0.34	0.02	0.32
Mn/TiO <sub>2</sub> -I	50	0.41	0.03	0.38
Fe/TiO <sub>2</sub> -I	47	0.34	0.03	0.31
TiO <sub>2</sub> -SG	131	0.28	0.07	0.21
Cu/TiO <sub>2</sub> -SG	103	0.21	0.05	0.16
Mn/TiO <sub>2</sub> -SG	125	0.19	0.07	0.12
Fe/TiO <sub>2</sub> -SG	93	0.14	0.06	0.09

**Table 2 – Phase composition, TiO<sub>2</sub> average crystallite size and lattice parameters (a and c) of the unit cell of all the catalysts.**

Sample	Anatase				Rutile			
	%	L (nm)	a (nm)	c (nm)	%	L (nm)	a (nm)	c (nm)
TiO <sub>2</sub> -P25	83	25	0.3787(3)	0.9521(3)	17	41	0.4597(1)	0.2968(1)
Cu/TiO <sub>2</sub> -I	84	25	0.3764(3)	0.9508(3)	16	31	0.4595(1)	0.2959(1)
Mn/TiO <sub>2</sub> -I	83	25	0.3780(3)	0.9504(3)	17	25	0.4587(1)	0.2981(1)
Fe/TiO <sub>2</sub> -I	80	25	0.3766(2)	0.9482(2)	20	41	0.4576(1)	0.3029(1)
TiO <sub>2</sub> -SG	100	9	0.3765(2)	0.9489(2)	–	–	–	–
Cu/TiO <sub>2</sub> -SG	100	10	0.3791(2)	0.9490(2)	–	–	–	–
Mn/TiO <sub>2</sub> -SG	100	7	0.3801(1)	0.9636(1)	–	–	–	–
Fe/TiO <sub>2</sub> -SG	100	7	0.3793(1)	0.9507(1)	–	–	–	–

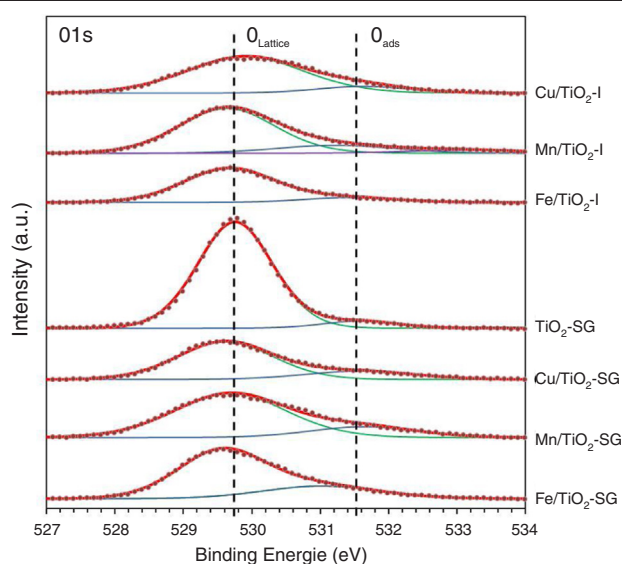
**Fig. 2 – Raman spectra of all the catalysts.**

#### Raman spectroscopy

Fig. 2 shows the Raman spectra of the different samples. All the catalysts exhibit four Raman bands with peaks, approximately, at 145, 390, 510 and 640 cm<sup>-1</sup>. According to the literature, the first one which is the most intense is assigned to the E<sub>g</sub> of an O–Ti–O type bending vibration while the other bands are related to B<sub>1g</sub>, (A<sub>1g</sub> + B<sub>1g</sub>) and E<sub>g</sub> modes, respectively [21,25,26]. Additional Raman band which appear as broad and poorly defined shoulder can be seen in some spectra at 200 cm<sup>-1</sup> that can be attributed to E<sub>g</sub> anatase mode [23]. Compared to pure TiO<sub>2</sub>, the most intense E<sub>g</sub> Raman peak at 145 cm<sup>-1</sup> for all the modified catalysts exhibits a shift toward higher wavenumbers and also suffers a change in its symmetry and broadness, this behavior is clearer in SG samples. This could indicate that the titania lattice is modified after the addition of transition metals in agreement with XRD results and suggest the formation of a structure of mixed oxides created probably by the substitution of Ti<sup>4+</sup> by Cu, Fe and Mn cations in titania lattice in a greater extent in SG samples [21,23].

#### X-ray photoelectron spectroscopy

Surface characterization of the samples was carried out by XPS analyses. The XPS spectra of O1s are shown in Fig. 3, which can be fitted by three peaks: lattice oxygen O<sub>Lattice</sub> (529.56–529.9 eV), chemisorbed oxygen O<sub>ads</sub> (531.02–531.63 eV) and hydroxyl groups (533.14 eV) due to moisture [27–29]. Moreover, the concentrations of O<sub>Lattice</sub> and O<sub>ads</sub> on the surface of each catalyst sample are calculated and presented in Table 3

**Fig. 3 – O1s XPS peaks of all the prepared catalysts.**

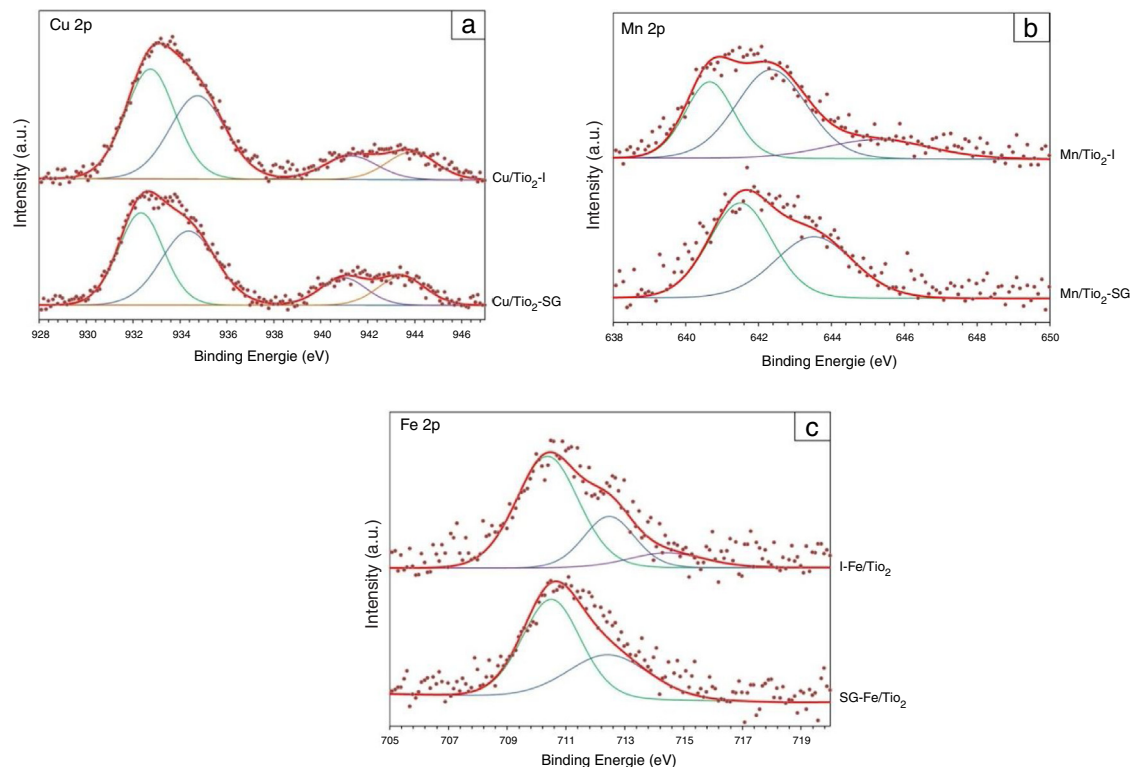
[30]. According to these results, there is a sharp increase in the percentage of chemisorbed oxygen on the surface of TiO<sub>2</sub> catalyst after the modification by oxides, in a greater extent in SG samples, which clearly depends to the oxygen vacancies generated by the introduction of metallic oxides. Indeed, the relative surface oxygen composition plays a key role in the catalytic activity. Where, O<sub>ads</sub> species have higher mobility than lattice oxygen, and some authors correlate the higher amount of HO<sup>-</sup> groups and oxygen vacancies with the higher photoactivity of a particular catalyst [31–33].

Fig. 4(a) shows the Cu 2p XPS spectra for Cu/TiO<sub>2</sub>-I and Cu/TiO<sub>2</sub>-SG. The shape and energy position of the two photoelectron peaks are nearly the same in the catalysts. The Cu 2p spectra of samples show four peaks, that localized at 932.73 and 932.33 eV indicates that the oxidation state of Cu present on the surface of the both samples is +1, the peak localized at 934.74 and 934.36 eV and the two satellite peaks at around 941.00 and 943.30 eV can be attributed to the presence of Cu<sup>2+</sup> [34–36].

The XPS spectra (Fig. 4(b)) of the Mn/TiO<sub>2</sub> sample elaborated by the impregnation method can be deconvoluted into three peaks at 640.65 eV, 642.35 eV and 645.27 eV corresponding to Mn<sup>2+</sup>, Mn<sup>4+</sup> and satellite peak, respectively. While, the sample

**Table 3 – XPS experimental results of all catalysts.**

Sample	O <sub>Lattice</sub>		O <sub>ads</sub>		Metal (a)		Metal (b)		Metal/Ti		
	BE (eV)	%	BE (eV)	%	BE (eV)	%	BE (eV)	%			
TiO <sub>2</sub> -P25 [25]	530.64	89.2	531.86	10.8	–	–	–	–	–		
Cu/TiO <sub>2</sub> -I	529.9	87.3	531.62	12.7	932.73	Cu <sup>+</sup>	42.88	934.74	Cu <sup>2+</sup>	36.43	0.185
Mn/TiO <sub>2</sub> -I	529.64	76.7	531.14	16.63	640.65	Mn <sup>2+</sup>	31.19	642.35	Mn <sup>4+</sup>	52.24	0.088
Fe/TiO <sub>2</sub> -I	529.68	87.3	531.44	12.4	710.36	Fe <sup>2+</sup>	65.36	712.47	Fe <sup>3+</sup>	24.4	0.091
TiO <sub>2</sub> -SG	529.75	93.65	531.59	6.35	–	–	–	–	–	–	
Cu/TiO <sub>2</sub> -SG	529.61	79.46	531.58	20.54	932.33	Cu <sup>+</sup>	38.16	934.36	Cu <sup>2+</sup>	37.71	0.182
Mn/TiO <sub>2</sub> -SG	529.68	81.15	531.63	18.85	641.5	Mn <sup>3+</sup>	55.55	643.52	Mn <sup>4+</sup>	44.45	0.068
Fe/TiO <sub>2</sub> -SG	529.56	73.48	531.02	26.52	710.5	Fe <sup>2+</sup>	61.62	712.44	Fe <sup>3+</sup>	38.38	0.051



**Fig. 4 – (a) Cu2p XPS peaks of Cu/TiO<sub>2</sub>-I and Cu/TiO<sub>2</sub>-SG. (b) Mn2p XPS peaks of Mn/TiO<sub>2</sub>-I and Mn/TiO<sub>2</sub>-SG. (c) Fe2p XPS peaks of Fe/TiO<sub>2</sub>-I and Fe/TiO<sub>2</sub>-SG.**

prepared by sol-gel method presents two peaks at 641.5 eV and 643.52 eV indicating the presence of Mn<sup>3+</sup> and Mn<sup>4+</sup>, respectively [31,37–39].

Fig. 4(c) shows that the Fe 2p curve of the both samples of Fe/TiO<sub>2</sub> is formed by two peaks located at around 710.4 and 712.4 eV can be attributed to the presence of Fe<sup>2+</sup> and Fe<sup>3+</sup> species, respectively [40,41].

The percentages of the Cu/Ti, Mn/Ti and Fe/Ti atomic ratios are reported in Table 3. In general, it can be observed that the catalysts elaborated by the sol-gel method present an increase in the atomic ratios compared to those elaborated by impregnation, mainly in the both samples of Mn/TiO<sub>2</sub> and Fe/TiO<sub>2</sub>, which can be considered as an estimation of metal dispersion on the support. Thus, these results show that the metal oxide particles are highly dispersed on the surface of the catalysts prepared by the impregnation process than those prepared by sol-gel, which may be due to the fact that, as suggested by

XRD and Raman results, the metallic cations are incorporated in the titania lattice in greater extend in SG samples while remain dispersed on the surface in impregnated samples.

#### UV-vis spectroscopy

As shown in Fig. 5, the both samples of TiO<sub>2</sub> reveal a sharp adsorption band at 350 nm ( $E_g = 3.1$  eV) associated to charge transference band O<sup>2-</sup>-Ti<sup>4+</sup>. While, in the modified samples additional absorption bands associated to the doping elements are detected and modify the band gap of the system (Table 4) measured considering the intense UV-vis charge transference band:

- (a) for Mn/TiO<sub>2</sub>-SG sample an intense band at 440 nm associated probably to O<sup>2-</sup>-Mn<sup>2+</sup> charge transference band (band gap 2.4 eV) and a moderate absorption band centered at 620 nm that can be associated to <sup>6</sup>A<sub>1g</sub>(S) → <sup>4</sup>T<sub>1g</sub>(G) of Mn<sup>2+</sup>

**Table 4 – Energy gap ( $E_g$ ) values of all catalysts.**

Sample	$E_g$ (eV)
TiO <sub>2</sub> -P25	3.1
Cu/TiO <sub>2</sub> -I	3.0
Mn/TiO <sub>2</sub> -I	3.0
Fe/TiO <sub>2</sub> -I	2.0
TiO <sub>2</sub> -SG	3.1
Cu/TiO <sub>2</sub> -SG	2.1
Mn/TiO <sub>2</sub> -SG	2.4
Fe/TiO <sub>2</sub> -SG	1.8

ions in octahedral symmetry are detected [42,43]. In the case of Mn/TiO<sub>2</sub>-I sample, both charge transference band and d–d transition bands shift to lower wavelength: 360 nm (band gap 3.0 eV) and 550 nm respectively, indicating a more intense crystal field in the impregnation samples.

(b) for Cu/TiO<sub>2</sub>-SG sample a sharp charge transference band associated to O<sup>2-</sup>–Cu<sup>2+</sup> is detected centered at 360 nm (band gap 2.1 eV) and an intense band at 780 nm associated to <sup>2</sup>B<sub>1g</sub> → <sup>2</sup>B<sub>2g</sub> due to an octahedral coordination with a strong tetragonal distortion of the Cu<sup>2+</sup> ions [42]. In the Cu/TiO<sub>2</sub>-I sample these bands shift to lower wavelength: 360 nm (band gap 3.0 eV) and 760 nm respectively, indicating a more intense crystal field in the impregnation samples as in the Mn samples case.

(c) for Fe/TiO<sub>2</sub>-SG sample an intense band at 420 nm (band gap 2.6 eV) associated to <sup>6</sup>A<sub>1g</sub>(S) → <sup>4</sup>A<sub>2g</sub>, <sup>4</sup>E(G) transition and a shoulder at 520 nm due to <sup>6</sup>A<sub>1g</sub>(S) → <sup>4</sup>T<sub>2g</sub>(G) both due to Fe<sup>3+</sup> in octahedral coordination [44]. However, for Fe/TiO<sub>2</sub>-I sample the charge transference band of O<sup>2-</sup>–Ti<sup>4+</sup> is detected at 350 nm and two shoulders at 420 nm and 520 nm associated to the above mentioned transition of Fe<sup>3+</sup> in octahedral coordination. In this case the continuous absorbance in the 300–540 nm range moves the calculation of the band gap to the absorption centered

at 520 nm (band gap of 1.8 and 2.0 eV for Fe/TiO<sub>2</sub>-SG and Fe/TiO<sub>2</sub>-I respectively).

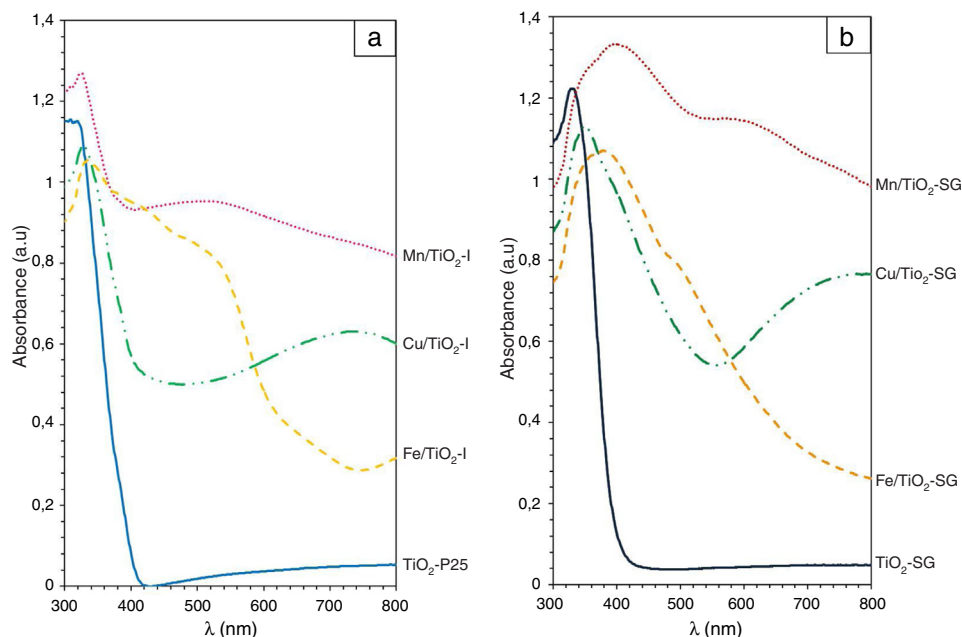
In all modified samples the band gap decreases associated to the red shift of the absorption associated to charge transference band of O<sup>2-</sup>–Ti<sup>4+</sup> that is modified by additional transference bands associated to doping cations. The values for the impregnation samples are close to the unmodified sample for Mn and Cu cases (3.0 eV versus 3.1 eV) but for sol-gel and Fe/TiO<sub>2</sub>-I samples the decrease is significant; all values are lower than 2.5 eV in agreement with a great extension of the entrance of cations in solid solution in this case.

#### Photocatalytic activity measurement

As has been mentioned above, the photocatalytic activity of the prepared catalysts has been evaluated in Orange II degradation in aqueous solution. Fig. 6 plots kinetic behavior of Orange II degradation for all samples. The kinetic parameters of the photocatalytic reaction and degradation rate are presented in Fig. 7. The degradation half time  $t_{1/2}$  was calculated using the Langmuir–Hinshelwood model as described elsewhere [44,45]. The photocatalytic activity of samples decrease in the order (TiO<sub>2</sub>-P25 > TiO<sub>2</sub>-SG) > (Mn/TiO<sub>2</sub>-I > Mn/TiO<sub>2</sub>-SG) > Fe/TiO<sub>2</sub>-SG > (Cu/TiO<sub>2</sub>-I > Cu/TiO<sub>2</sub>-SG) > Fe/TiO<sub>2</sub>-I.

The results show that nearly all samples own a degradation rate superior to 90%, except in the case of the both catalysts Fe/TiO<sub>2</sub>-I (69%) and Cu/TiO<sub>2</sub>-SG (82%) associated to a  $E_g$  of 2.0 and 2.1 eV respectively. The high photoactive samples show relative high band gap (3–3.1 eV) except for Cu/TiO<sub>2</sub>-I sample and the lower show relative low band gap (2–2.1 eV) except for Fe/TiO<sub>2</sub>-SG.

In all cases the SG samples show lower photoactivity than its homologous I samples; this behavior can be explained by the presence of both anatase and rutile phases in the commercial TiO<sub>2</sub>. Indeed, the phase junction formed between



**Fig. 5 – The UV-vis spectra of TiO<sub>2</sub>-P25 and the catalysts prepared by impregnation (a) and sol-gel (b).**

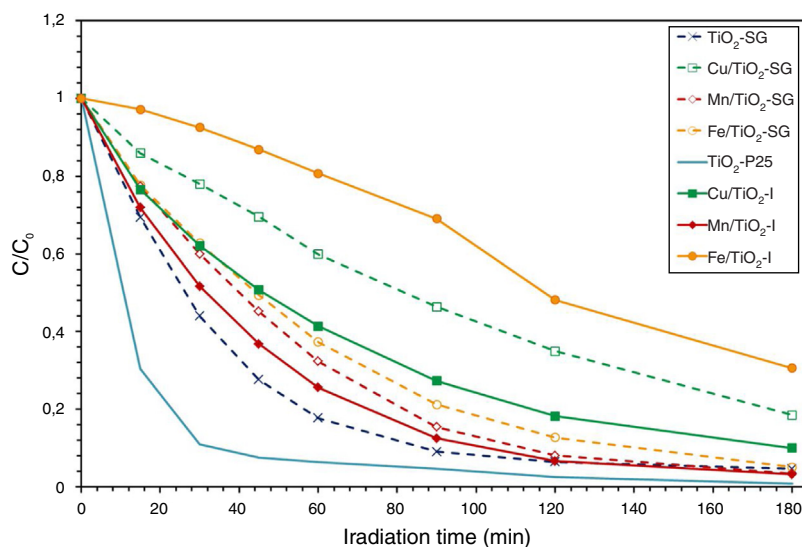


Fig. 6 – Kinetic behavior of Orange II degradation by  $\text{TiO}_2$ -P25 and the prepared catalysts.

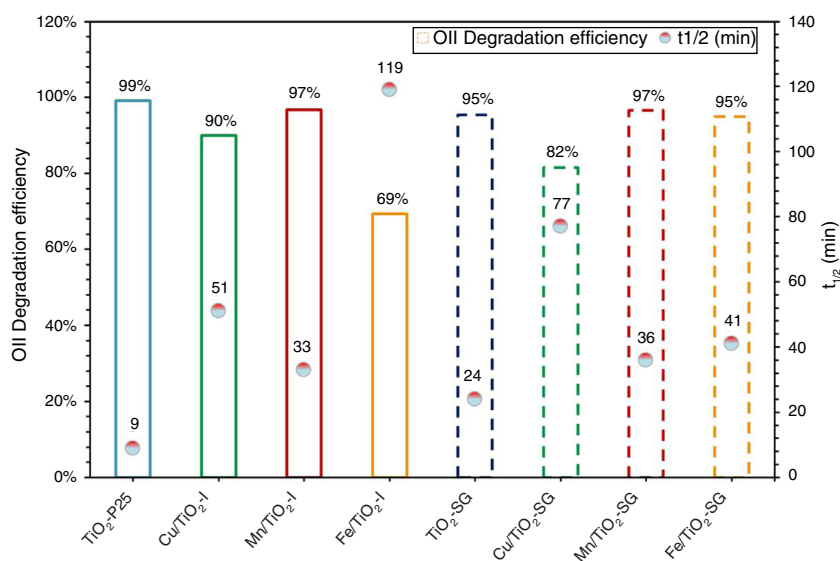


Fig. 7 – Degradation rate of Orange II by  $\text{TiO}_2$ -P25 and the prepared catalysts.

the surface anatase nanoparticles and rutile particles may facilitate transfer of the photogenerated electron from the conduction band of the rutile phase to the trapping sites on the anatase surface, this electron mobility prevents the electron-hole recombination and allows the holes generated to move to the surface of the catalyst [46–50].

As above has been discussed, XRD, Raman, XPS and UV-vis data show that the metallic cations are incorporated in the titania lattice in greater extent in SG samples while remain dispersed on the surface in impregnated samples. Although, it is well known that the entrance of metallic cations in solid solution associated to red-shift in the absorption band edge, and the increase in absorption intensity enhance the photocatalytic activity due to the enhanced formation

of photoelectrons and photoholes [51,52], at a high doping content, a large number of structural defects could be induced serving as a recombination center of electron-hole photogenerated, which absolutely slow down the photocatalytic process [32,52–54]. Therefore, the SG samples show higher photoactivity than I samples except in the case of Fe modified samples; in this case, the red shift is very large ( $E_g = 1.8\text{ eV}$  is the lower value observed) and then compensate the induced recombination center of electron-hole pair. In effect, the minimum discoloration is reached at 69.4% for iron modified titania and elaborated by impregnation method. Whereas the sample prepared by sol-gel method presents 95%, which can be related to the simultaneous action of three different factors: the preservation of the



charge transference band of  $O^{2-}-Ti^{4+}$  detected at 350 nm in the Fe/TiO<sub>2</sub>-I sample that inhibits the semiconductor mechanism of  $O^{2-}-Ti^{4+}$  in favor of d–d transition associated to iron ions, the high surface area and the higher rate chemisorbed oxygen found in Fe/TiO<sub>2</sub>-SG sample, that can inhibit the recombination hole–electron due to the high content of iron.

On the other hand, the Cu/TiO<sub>2</sub> samples present a decrease in the photocatalytic activity compared to unmodified TiO<sub>2</sub> due to the high copper content. Indeed, the excessive oxygen vacancies and Cu species can become the recombination centers of the photogenerated electron–hole [32]. Furthermore, the difference between the degradation rates of samples may be explained by the presence of the both phase anatase and rutile in the sample elaborated by impregnation method may slow down the electron–hole recombination [50,55].

According to all obtained results, the best photocatalytic activity was observed for TiO<sub>2</sub> prepared by a sol–gel method which presented a half-life time of 24 min nearest to TiO<sub>2</sub>-P25 (9 min). The Mn modified samples show slightly higher half-life time (33–36 min), the introduction of Fe and Cu ions in the lattice increases the half-life time. All this indicates that to obtain a highly active catalyst it necessary to optimize four properties of the samples: the rutile phase content, the doping content, the chemisorbed oxygen content and the high surface area.

## Conclusion

In this study, we have successfully prepared Fe, Mn and Cu modified TiO<sub>2</sub> by impregnation and sol–gel methods followed by calcination process. The prepared samples have been characterized by XRD, Raman, physical adsorption of N<sub>2</sub> at 196 °C, XPS and UV–vis and investigated through photocatalytic degradation of azoic dye “Orange II”.

The characterization results showed the presence of the both phase anatase rutile for the samples prepared by impregnation method. On the other hand, no metallic oxides diffraction peaks were observed, suggesting a good dispersion of the metal species on the catalyst surface. This result was also confirmed by the Metal/Ti ratio by XPS analyses. The modification of TiO<sub>2</sub> decreases the BET surface area of samples and reduces the band gap energy.

The results show that nearly all samples own a degradation rate superior to 90%, except in the case of the both catalysts Fe/TiO<sub>2</sub>-I (69%) and Cu/TiO<sub>2</sub>-SG (82%) associated to a  $E_g$  of 2.0 and 2.1 eV respectively. The high photoactive samples show relative high band gap (3–3.1 eV) except for Cu/TiO<sub>2</sub>-I sample and the lower show relative low band gap (2–2.1 eV) except for Fe/TiO<sub>2</sub>-SG.

In all cases the SG samples show lower photoactivity than its homologous I samples associated to the presence of both anatase and rutile phases in the commercial TiO<sub>2</sub>; the phase junction formed between anatase–rutile particles facilitate transfer of the photogenerated electron from the conduction band of the rutile phase to the trapping sites on the anatase surface and prevents the electron–hole recombination and allows the holes generated to move to the surface of the catalyst.

XRD, Raman, XPS and UV–vis data show that the metallic cations are incorporated in the titania lattice in greater extend in SG samples while remain dispersed on the surface in impregnated samples. Although the entrance of metallic cations in solid solution enhance the photocatalytic activity due to the enhanced formation of photoelectrons and photoholes, at a high doping content, a large number of structural defects could be induced serving as a recombination center of electron–hole photogenerated, therefore the SG samples show higher photoactivity than I samples except in the case of Fe modified samples; in this case, the red shift is very large ( $E_g = 1.8$  eV is the lower value observed) and then compensate the induced recombination center of electron–hole pair.

The best photocatalytic activity was observed for TiO<sub>2</sub> prepared by a sol–gel method, which presented a half-life time of 24 min nearest to TiO<sub>2</sub>-P25 (9 min). The Mn modified samples show slightly higher half-life time (33–36 min), the modification by Fe and Cu ions increases the half-time. All this indicates that to obtain a highly active catalyst it necessary to optimize four properties of the samples: the rutile phase content, the doping content, the chemisorbed oxygen content and the high surface area.

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